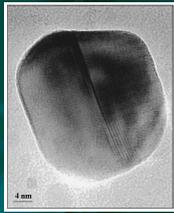


## Diamond formation from graphite under highly charged ion irradiation

## $sp^3$ formation by high dose implantation

- Transformation of  $sp^2$  to  $sp^3$  under high dose implantation (formation of DLC).
- High dose ion implantation induces damages into graphite.

## Nanodiamond formation by swift ion bombardment into graphite



- Found in the acid residue from 350 MeV Kr ion irradiated graphite
- Room temperature
- Low yield (~ 0.01/ion)

T. L. Daulton et al., Nucl. Instr. and Meth. in Phys. Res. B175-177, 12 (2001)

## Unique features of HCI

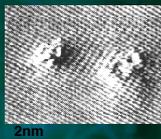
- Large potential energy
- Energy transfer occurs before the direct collision
- Multiple electron emission occurs from the small area for short period.
- Local modification of surface electronic states

## HCI irradiation on HOPG

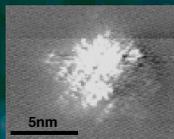
Ar<sup>8+</sup> : 4 keV

Xe<sup>44+</sup> : 276 keV

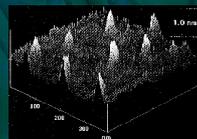
U<sup>79+</sup> : 2.2 keV/amu



K. Mochiji et al. J. Appl. Phys. 82, 6037 (1997).

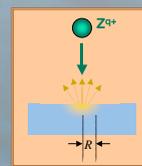


R. Mirmitt, et al. Phys. Scr., T92, 22 (2001)



C. Uehiroke et al. Nucl. Instr. Meth. In Phys. Res. B90, 528 (1995)

## Maximum radius of electron emission



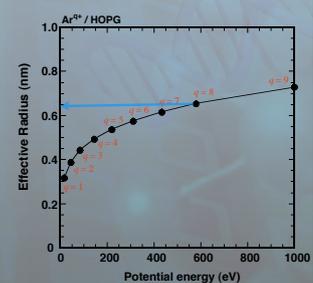
$$R = (\epsilon W_q / 0.08 \pi n^2 e^2)^{1/5}$$

$\epsilon$  : the dielectric constant of the solid

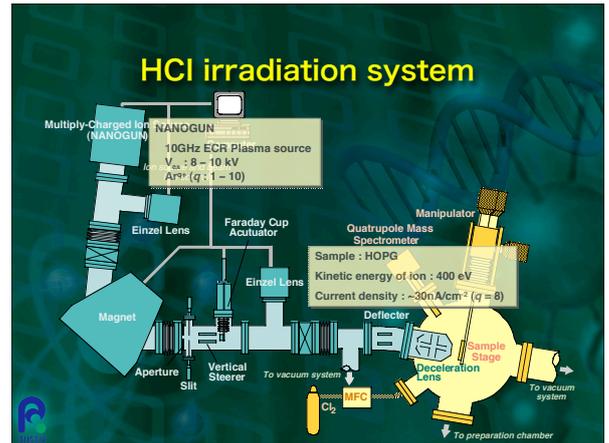
$W_q$  : the total potential energy of a  $q$ -charge state ion

$n$  : the atomic density of the solid

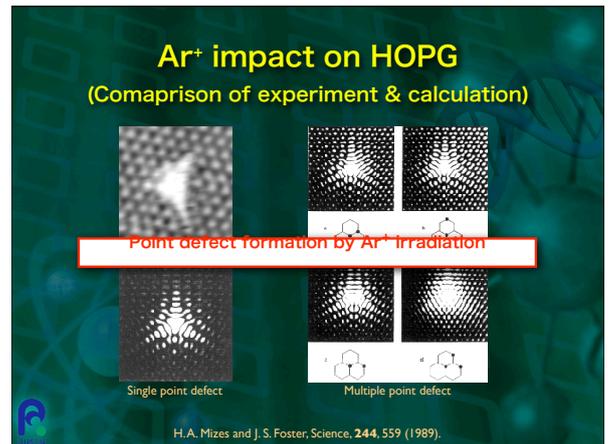
$e$  : the electron charge



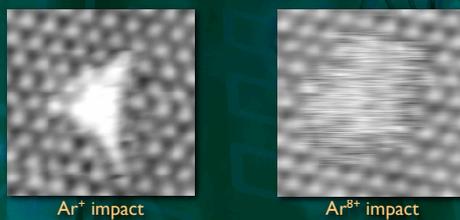
## Experimental



## Slow $\text{Ar}^{8+}$ impact on HOPG



## Comparison of STM images after irradiation of $\text{Ar}^+$ and $\text{Ar}^{8+}$

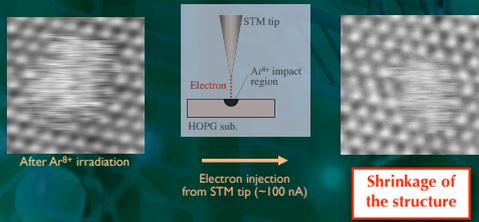


Complex structure is formed by  $\text{Ar}^{8+}$  irradiation

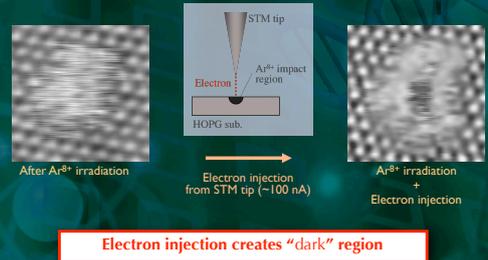
## Slow $\text{Ar}^{8+}$ impact on HOPG

Electron injection into the  $\text{Ar}^{8+}$  impact region

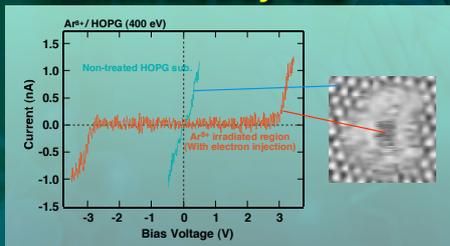
### Electron injection to Ar<sup>8+</sup> impact region



### Electron injection to Ar<sup>8+</sup> impact region

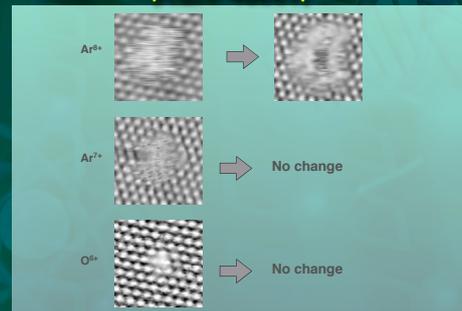


### I-V characteristic of Ar<sup>8+</sup> impact area measured by STS

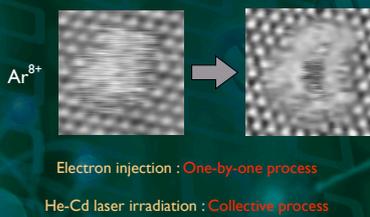


Metallic graphite changes to the non-conductive by electron injection

### Comparison of STM images of HOPG surface after Ar<sup>8+</sup> irradiation (K.E. = 400eV)



### Transition from metallic to non-conductive



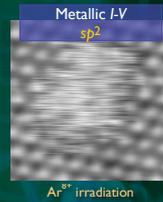
### Slow Ar<sup>8+</sup> impact on HOPG

Hydrogen treatment

## Why the hydrogen treatment ?

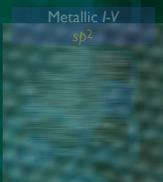
- Hydrogenated  $sp^3$  rich amorphous carbon
  - Blue emission in PL
- Hydrogen-terminated diamond surface
  - Negative electron affinity

## Evolution of STM images of HOPG by $Ar^{8+}$ impact and post processes



$Ar^{8+}$  irradiation

## Evolution of STM images of HOPG by $Ar^{8+}$ impact and post processes



$Ar^{8+}$  irradiation



$Ar^{8+}$  irradiation +  
Electron injection  
(He-Cd laser irradiation)

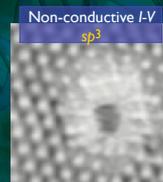
## Evolution of STM images of HOPG by $Ar^{8+}$ impact and post processes



$Ar^{8+}$  irradiation

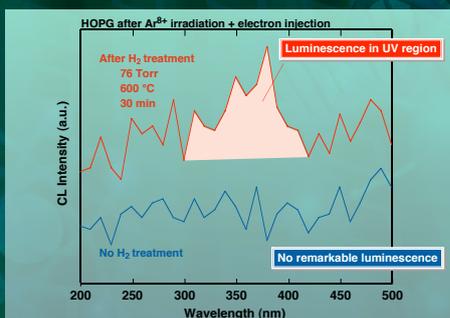


$Ar^{8+}$  irradiation +  
Electron injection  
(He-Cd laser irradiation)

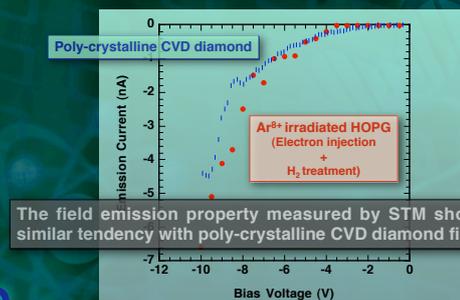


$Ar^{8+}$  irradiation +  
Electron injection  
(He-Cd laser irradiation) +  
 $H_2$  treatment  
(76 Torr, 600 °C, 30 min)

## Cathode Luminescence (@20K)



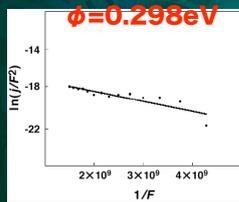
## Field emission property measured by STM



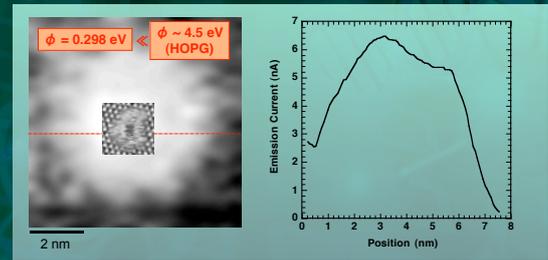
## Work function of the Ar<sup>8+</sup> impact region after electron injection and hydrogen treatment

$$j = \frac{e^3 F^2}{8\pi h \phi^2 ((e^3 F)^{1/2} / \phi)} \times \exp \left[ \frac{-4(2m)^{1/2} \phi^{3/2}}{3heF} v((e^3 F)^{1/2} / \phi) \right] \frac{\pi kT / d}{\sin(\pi kT / d)}$$

$e$  : the elementary electric charge  
 $F$  : the electric field intensity  
 $h$  : Planck's constant  
 $\phi$  : the work function  
 $m$  : the rest electric mass  
 $j$  : the current density  
 $d$  : the diameter of the emission area



## Field emission properties of Ar<sup>8+</sup> impact region after electron injection and H<sub>2</sub> treatment



## Effects of Ar<sup>8+</sup> irradiation on graphite with post treatments

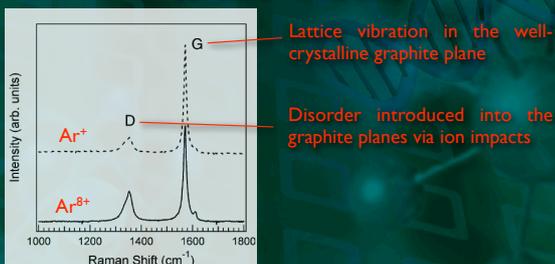
- Metallic → Non-conductive ( $E_g = -6 \text{ eV}$ )
- Field emission property similar to the CVD-diamond film ( $\phi = 0.298 \text{ eV}$ )

Transition from  $sp^2$  to  $sp^3$  occurs.

## Slow Ar<sup>8+</sup> impact on HOPG

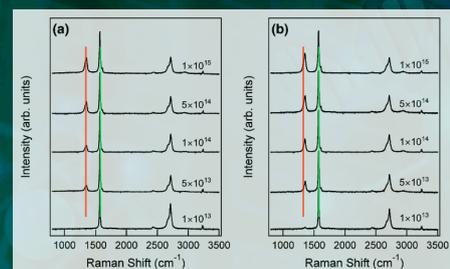
Plausible formed structure

## First-order Raman spectra of HOPG irradiated with Ar<sup>+</sup> and Ar<sup>8+</sup> (@ $5 \times 10^{13} \text{ ions/cm}^2$ )



$I(D)/I(G)$  for Ar<sup>8+</sup> is larger than that for Ar<sup>+</sup> irradiation

## Comparison of Raman spectra HOPG irradiated with Ar<sup>+</sup> and Ar<sup>8+</sup>

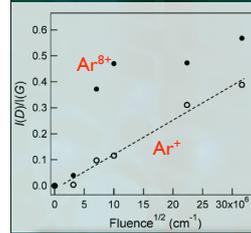


$I(D)/I(G)$  increases with the ion fluence

## Relation between G and D peaks

- $I(D)/I(G)$  is inversely proportional to the in-plane crystallite size  $L$  of graphite.
- $L$  can be regarded as equal to the mean distance between vacancies introduced by the single impact of singly-charged ion.
- $I(D)/I(G)$  is proportional to the square root of ion fluence  $\phi^{1/2}$ .

## $I(D)/I(G)$ vs $\phi^{1/2}$ for the irradiation of Ar and Ar<sup>8+</sup>

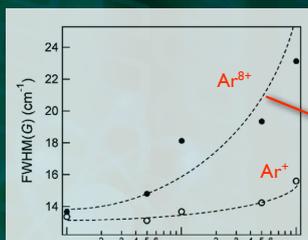


For Ar<sup>+</sup> irradiation,  $I(D)/I(G)$  increases linearly with the increasing of  $\phi^{1/2}$ .

Ar<sup>+</sup> irradiation induces the vacancy formation.

The defects introduced by Ar<sup>8+</sup> irradiation are NOT simple vacancies BUT some defect complex.

## FWHM(G) vs ion fluence



FWHM(G) increases when the clustering of vacancies starts in graphite.

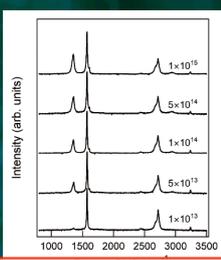
FWHM(G) increases rapidly for Ar<sup>8+</sup> irradiation.

The irradiation-induced defects for Ar<sup>8+</sup> must be vacancy clusters.

## Formation of vacancy clusters (singly charged ion)

- In case of Ar<sup>+</sup> irradiation, the formation of the vacancy clusters is followed by the amorphization of graphite, because the clustering occurs as a result of the overlap of the cascades for the direction parallel to the surface.

## Formation of vacancy clusters (Highly charged ion)



Enough intensity  
Sharp peak

Amorphization does not occur for Ar<sup>8+</sup> irradiation

## Summary of Raman spectroscopic measurements

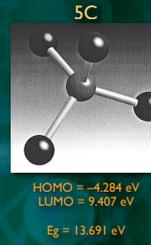
- The defects introduced by Ar<sup>8+</sup> impacts is NOT simple vacancies as is the case of Ar<sup>+</sup> impacts but vacancy clusters.
- Second-order Raman features retained enough intensity and sharpness against Ar<sup>8+</sup> irradiation in spite of the formation of vacancy clusters which leads to the amorphization for Ar<sup>+</sup> irradiation.

## Summary of Raman spectroscopic measurements

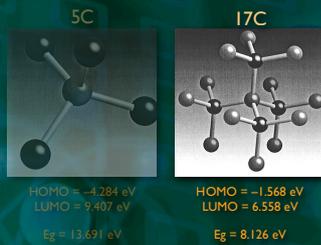
- The defects introduced by  $\text{Ar}^{8+}$  impacts is **NOT** simple vacancies as is the case of  $\text{Ar}^+$  impacts but **vacancy clusters**.
- Second-order Raman features retained enough intensity and sharpness against  $\text{Ar}^{8+}$  irradiation in spite of the formation of vacancy clusters which leads to the amorphization for  $\text{Ar}^+$  irradiation.

Defects formation by  $\text{Ar}^{8+}$

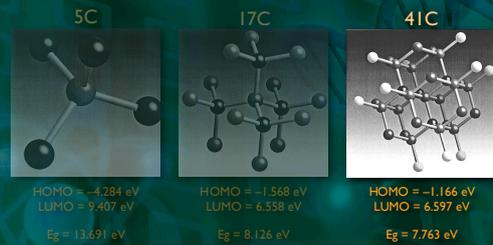
## Energy gap vs number of carbon atoms



## Energy gap vs number of carbon atoms

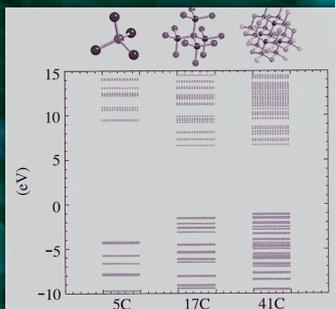


## Energy gap vs number of carbon atoms



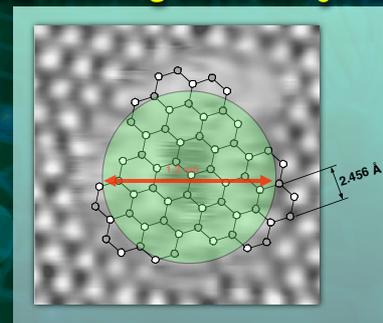
Symmetric diamond structures show **LARGE** energy gap

## Energy gap vs number of carbon atoms



T. Ishii (Kagawa Univ.), Private Communication.

## Complex structure formed by $\text{Ar}^{8+}$ impact and the following electron injection



### sp<sup>3</sup> cluster models with 20 carbon atoms



**spiral**  
12.240 eV



**round**  
8.983 eV



**hanger**  
5.948 eV



**zigzag**  
11.444 eV

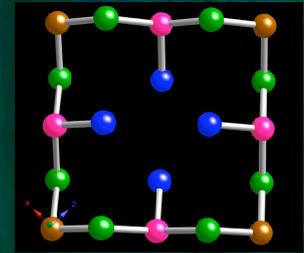


**C3v**  
8.237 eV

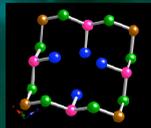
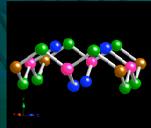
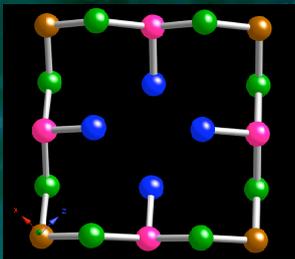


**fly**  
7.107 eV

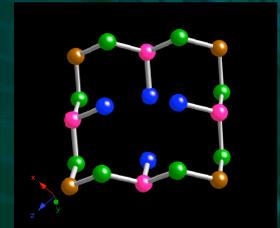
### Band gap of S4 cluster



### C<sub>20</sub> cluster (S4)

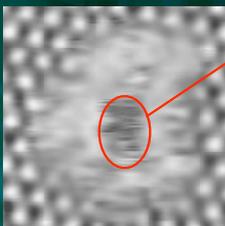


### S4 cluster

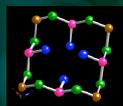


Defects formation by Ar<sup>8+</sup>

### Structure of HCl impact region

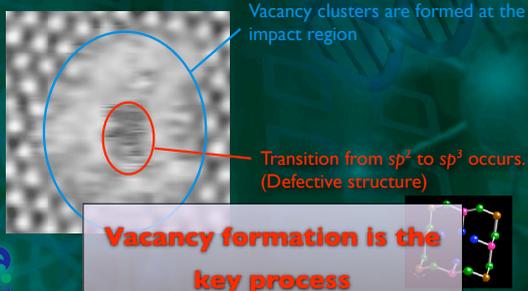


Transition from sp<sup>2</sup> to sp<sup>3</sup> occurs.



### Plausible structure of the impact region

## Structure of HCl impact region



## Why the transition from $sp^2$ to $sp^3$ occurs?

## From graphite to diamond ?

- From an *ab-initio* calculation for the core-excited states in diamond and graphite, the transition barrier from graphite to diamond becomes zero with core excitation or doping of multiple holes in graphite.

Univ. of Tokyo/Prof. Watanabe,  
Osaka Univ./Prof. Yoshida

## Multiple holes generated by HCl induces the transition from graphite to diamond?

- Generation of multiple holes at the graphite surface by approach of HCl
- Multiple hole generation reduces the transition barrier from graphite to diamond
- Transition from  $sp^2$  to  $sp^3$  occurs

## When the slow HCl approaches to the graphite surface,

Large Coulomb potential of HCl induces ...

- ... the generation of multiple holes,  
Reduction of the transition barrier from graphite to diamond.
- ... and, the formation of vacancy (cluster).  
Generation of the low band gap structure.

## Why the electron injection process is necessary to form nanodiamond ?

- Direct collision of ion affects the nano-diamond structure ?

## New approach

Suppress the effect of direct collision.

(Low K.E. condition)



Extremely slow HCl

## Generation of extremely slow HCl

(collaboration with Prof. Kumagai & Dr. Matsuo)

Atom cooling by MO trap

+

Laser ionization by high power laser

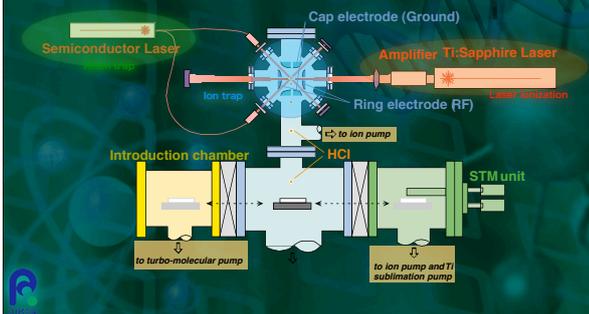
+

Ion capture by RF trap

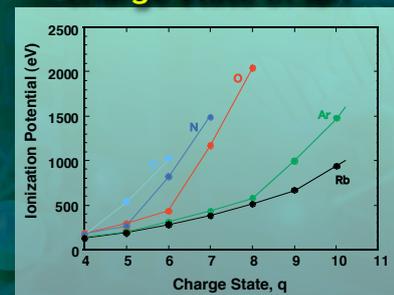
+

In situ STM observation

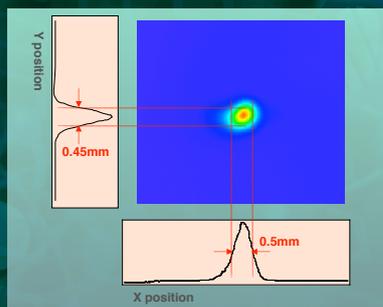
## Extremely slow HCl generation using multi-trap system



## Ionization potential vs charge state of ion

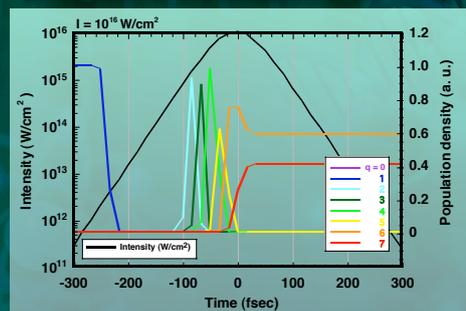


## Rb atoms in the MO trap system

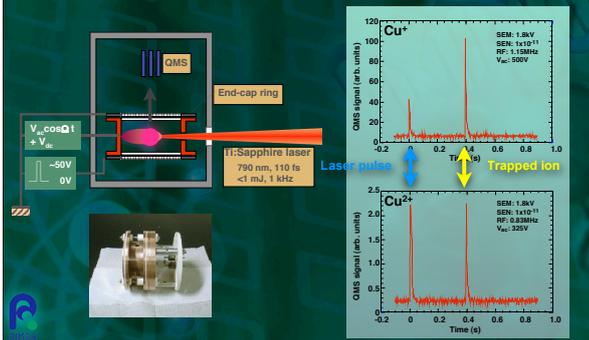


## Laser ionization

(Simulation of the ionization of Rb atom)



## Ion capture by RF trap system



## Summary

- $\text{Ar}^{8+}$  impact and the electron injection creates non-conductive  $\text{sp}^3$  region on graphite.
- After the hydrogen treatment, the impact region shows the field emission tendency similar to the CVD diamond film.
- Band gap of S4 cluster shows a good agreement with the experimental results.
- From the Raman spectroscopy, vacancy clusters are formed by the  $\text{Ar}^{8+}$  impact.

### Collaboration

Dr. A. Hida (RIKEN (formerly Univ. of Tokyo))  
 Prof. Ishii (Kagawa Univ.)  
 Prof. Y. Aoyagi (Ritsumeikan Univ., (formerly RIKEN))

Prof. Y. Yamamoto (Hosei Univ.)  
 Prof. H. Takai (Tokyo Denki Univ.)  
 Prof. K. Maeda (Univ. of Tokyo)  
 Prof. T. Ishii (Kagawa Univ.)

Prof. H. Kumagai (Osaka Pref. Univ.)  
 Dr. Y. Matsuo (RIKEN)

### Great Thanks

Prof. Y. Yamazaki (Univ. of Tokyo, RIKEN)